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Fang, Y

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8 **Percolation threshold and effective properties of CNTs-reinforced two-phase composite**  
9 **materials**

10 Yuan Fang<sup>1</sup>, Shouwang Hu<sup>2</sup>, Long-yuan Li<sup>3\*</sup>, Sung-Hwan Jang<sup>4</sup>

11 1) Guangdong Provincial Key Laboratory of Durability for Marine Civil Engineering,  
12 Shenzhen University, Shenzhen 518060, P R China (yuanfang@szu.edu.cn)

13 2) School of Environment and Civil Engineering, Dongguan University of Technology,  
14 Dongguan 523808, P R China (shouwang.hu@dgut.edu.cn)

15 3) School of Engineering, Computing and Mathematics, University of Plymouth, Plymouth,  
16 Devon PL4 8AA, UK (long-yuan.li@plymouth.ac.uk)

17 4) School of Engineering, Hanyang University ERICA, Ansan, Gyeonggi-do 15588, South  
18 Korea (sj2527@hanyang.ac.kr)

19 **Abstract** – In this paper an analytical prediction model is developed to determine the  
20 percolation threshold and effective properties of carbon nanotubes reinforced two-phase  
21 composite materials. The model considers the effect of not only volume fraction but also aspect  
22 ratio and mixed pattern of fibres in the composites. It is shown that the effect of fibres on the  
23 effective properties of the composites can be characterised in terms of three different zones,  
24 namely non-percolated zone, partially percolated zone, and fully percolated zone. The  
25 formulations for determining percolation threshold and effective properties of the composites  
26 in the three different zones are derived. The model is validated by using available experimental  
27 data to demonstrate its appropriateness and reliability.

28 **Keywords:** Composites; CNTs; Percolation threshold; Effective properties; Aspect ratio.

29 \* Corresponding author

30 **1. Introduction**

31

32 A composite material is a combination of two or more materials with different physical and  
33 chemical properties. The bulk properties of the composite material can be significantly  
34 different from those of any of its individual constituents. By choosing an appropriate  
35 combination of constitutive materials, manufacturers can produce properties that closely fit the  
36 requirements for a particular structure with a particular purpose. Nowadays composite  
37 materials have been used in many fields such as aerospace, architecture, automotive, energy,  
38 infrastructure, marine, military, and sports and recreation.

39

40 It is well known that for a matrix-composite material its thermal, electrical and mechanical  
41 properties, often named as effective properties, are dependent on the properties of the matrix  
42 and fillers or inclusions forming the composite, the volume fraction and aspect ratio of the  
43 fillers, and the way how the fillers are dispersed and distributed in the matrix [1,2]. It has been  
44 widely reported that, the effective property of a composite, such as its electrical conductivity,  
45 changes slightly when the volume fraction of fillers is very small, but sharply when the volume  
46 fraction of fillers reaches to a certain narrow range [3]. This phenomenon is characterised as  
47 the percolation [4], which describes the connectivity of fillers within the matrix and its effect  
48 on the macroscale properties of the composite. In literature various percolation models have  
49 been developed to describe the effective properties of fibre-reinforced composite materials in  
50 which a sharp insulator–conductor transition is typically observed with increasing filler fraction  
51 [5]. For instance, Wu et al. [6] carried out an experimental investigation on the conductivity of  
52 asphalt concrete containing conductive fillers. Ma and Gao [7] developed a 3-D Monte Carlo  
53 model for predicting electrical conductivity of polymer matrix composites filled with  
54 conductive curved fibres. It was found that the curliness largely influenced the percolation

55 threshold, and the more curved the fibre, the higher the threshold. Savchenko and Ionov [8]  
56 used experimental methods to investigate the electrical and thermal properties of binary  
57 systems consisting of stearine, expanded and fine-crystalline graphite. It was demonstrated that  
58 the percolation threshold depended on the aspect ratio of the electro-conducting filler.  
59 Dubnikova et al. [9] examined the effects of multi-walled carbon nanotubes (CNTs)  
60 dimensions and surface modification on the morphology, mechanical reinforcement, and  
61 electrical properties of polypropylene-based composites prepared by melt mixing. It was  
62 demonstrated that the nanocomposites based on long large diameter CNTs had the lowest  
63 percolation threshold. Martone et al. [10] examined the effect of dispersed multi-walled CNTs  
64 on the bending modulus of their reinforced epoxy systems. It was shown that reinforcement  
65 efficiency was characterised by two limiting behaviours whose transition region coincides with  
66 the development of a percolative network of CNTs. Lonjon et al. [11] provided a comparative  
67 study on the percolation threshold of two polymer composites; one was reinforced with  
68 nanowires and the other was with spherical nanoparticles. Feng and Jiang [12] presented a  
69 mixed micromechanics model to predict the overall electrical conductivity of CNTs-reinforced  
70 polymer composites. It was shown that the size of CNTs had a significant effect on the  
71 percolation threshold of nanocomposites. Wang et al. [13] examined the effect of aspect ratio  
72 on the electrical conductivity and crystallization kinetics of semi-crystalline polystyrene  
73 composites filled with graphene nano-sheets. Kim et al. [14] presented an analytical  
74 homogenization approach for composites containing multiple heterogeneities with conductive  
75 coated layers to predict the percolation threshold of polymer composites containing randomly  
76 oriented ellipsoidal inclusions. Yang et al. [15] proposed a micromechanics-based model to  
77 investigate the effect of CNTs agglomeration on the electrical conductivity and percolation  
78 threshold of nanocomposites. Wu et al. [16] investigated the thermal and electrical properties  
79 of polypropylene composites with dense small-sized multi-walled CNTs network located

80 within loosened large-sized expanded graphite network. It was suggested that the formation of  
81 double percolated filler network could effectively reduce the interface thermal resistance.  
82 Taherian [17] presented an analytical formula to predict the electrical conductivity of  
83 composites reinforced by conductive fillers. The work employed a sigmoidal equation to  
84 describe the relationship between electrical conductivity and filler volume fraction. The results  
85 showed that the aspect ratio and size of fillers were the most important factors affecting  
86 percolation threshold. Nilsson et al. [18] developed three different simulation models based on  
87 finite element, percolation threshold and electrical networks, respectively for predicting  
88 electrical conductivity and percolation threshold of field-grading polymer composites. Chiu et  
89 al. [19] investigated the rheological and electrical properties of syndiotactic polystyrene  
90 composites filled with graphene nano-sheets and CNTs to reveal the effect of filler  
91 concentration, in which percolation scaling laws were applied to the magnitudes of storage  
92 modulus and electrical conductivity to determine the threshold concentration and  
93 corresponding exponent function. Lu et al. [20] proposed a numerical modelling framework to  
94 evaluate the effective electric conductivity in polymer composites reinforced with graphene  
95 sheets, taking into account the electrical tunnelling effect. Fang et al. [21] presented an  
96 analytical study on the electrical conductivity of composites whose constitutive materials have  
97 distinct electrical properties. Formulations were derived for determining percolation threshold  
98 and electrical conductivity of composites with considering aspect ratio effect. Zare and Rhee  
99 [22] examined the effects of filler network and interfacial shear strength on the mechanical  
100 properties of CNTs-reinforced nanocomposites. Xu et al [23] proposed a theoretical framework  
101 for predicting excluded volume and percolation threshold of soft interphase and effective  
102 conductivity of carbon fibrous composites. Recently, Tang et al. [24] presented an analytical  
103 model to predict the percolation threshold and electrical conductivity of CNTs-reinforced  
104 composites by taking into account the effects of CNTs waviness and dispersion. It was

105 indicated that high electrical conductivity could be achieved for composites with CNTs of high  
106 conductivity, large aspect ratio, perfect dispersion state and low degree of waviness. Chang et  
107 al. [25] developed a simulation model to examine the effect of deformation on the percolation  
108 threshold and electrical conductivity of composites with fillers, in which the percolating filler  
109 networks were modelled as an equivalent electrical circuit consisting of tunnelling and intrinsic  
110 resistances. Xu et al. [26] performed a numerical simulation to estimate the percolation  
111 threshold of 3-D porous media interacted by anisotropic-shaped pores and cracks and their  
112 effects on the thermal conductivity and elastic modulus of the media. Chang et al. [27]  
113 developed a numerical model on the structural evolution of conductive polymer composites  
114 subjected to mechanical loading action. The model was used to examine the effect of  
115 deformation on the percolation threshold and effective electrical conductivity of composites  
116 with fillers. Fang et al. [28] presented an analytical model which can be used to examine the  
117 effect of mechanical strains on the percolation threshold and effective electrical conductivity  
118 of CNTs-reinforced two-phase composites.

119

120 CNTs have excellent thermal, electrical, and mechanical properties. The use of CNTs as  
121 inclusions in polymer matrix can improve the thermal, electrical, and mechanical properties of  
122 the polymer-based composites and produce functionally graded smart materials and/or smart  
123 sensors. To achieve the purpose of the improvement and function required, however, it is  
124 necessary to know the percolation threshold of CNTs for a given CNTs-reinforced polymer  
125 composite. The literature survey described above shows that there have been numerous  
126 experimental and numerical investigations but limited prediction models available on the  
127 percolation threshold to elucidate how the effective properties vary with CNTs volume fraction.  
128 In this paper, an analytical model is developed to predict the percolation threshold and effective  
129 properties of fibre reinforced two-phase composite materials. The model considers the effect of

130 volume fraction, aspect ratio, and mixed pattern of fibres in the composites. Compared to  
 131 existing analytical models published in literature, the present model has the following new  
 132 features. Firstly, the percolation threshold is derived based on randomly dispersed fibres which  
 133 are characterised by three identical ellipsoids perpendicularly placed along with three axes in  
 134 Cartesian coordinate system. Secondly, the effective properties are calculated based on three  
 135 different zones, namely non-percolated zone, partially percolated zone, and fully percolated  
 136 zone. These three zones are characterised mathematically based on the aspect ratio and volume  
 137 fraction of inclusions in the composite. Thirdly, the prediction formulas derived can be applied  
 138 to all of the three zones. Fourthly, to demonstrate the present model, comparisons between the  
 139 present model and existing experimental data are also provided for composites with different  
 140 constitutive materials for various different volume fractions of inclusions in different zones.

141

## 142 **2. Effective properties of two-phase composite materials**

143

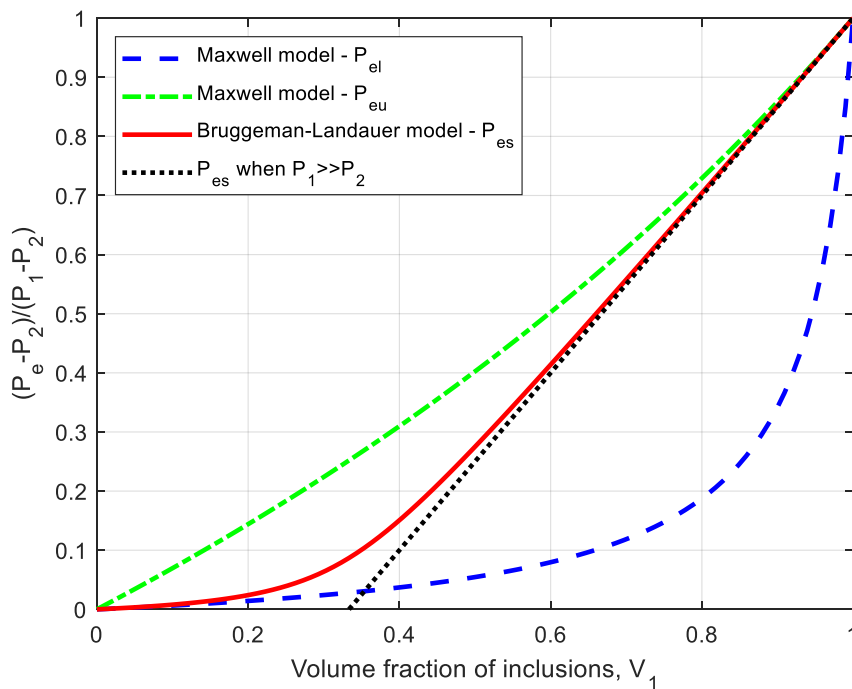
144 In a two-phase matrix composite material, the matrix and fillers can be treated as the  
 145 homogeneous medium and inclusions in the composite, respectively. The effective properties  
 146 of the two-phase composite such as the thermal conductivity, electrical conductivity, diffusion  
 147 coefficient, and elastic modulus, can be estimated using the Maxwell model [29] if the  
 148 inclusions are small spheres and perfectly enclosed by the matrix,

$$149 \quad P_{el} = P_2 \frac{(P_1 + 2P_2) + 2V_1(P_1 - P_2)}{(P_1 + 2P_2) - V_1(P_1 - P_2)} \quad (1)$$

150 where  $P_{el}$  is the effective property of the composite,  $V_1$  is the volume fraction of the inclusions,  
 151  $P_1$  and  $P_2$  are the properties of the inclusions and matrix material, respectively. Conversely, if  
 152 we assume the matrix material is made by small spheres that are perfectly enclosed by the  
 153 inclusions, then a different version of Eq.(1) can be obtained as follows,

$$154 \quad P_{eu} = P_1 \frac{(P_2 + 2P_1) + 2(1 - V_1)(P_2 - P_1)}{(P_2 + 2P_1) - (1 - V_1)(P_2 - P_1)} \quad (2)$$

155 where  $P_{eu}$  is also the effective property of the composite. For the case of  $P_1 > P_2$  Eqs.(1) and (2)  
 156 give the lower- and upper-bounds of the effective property of the two-phase composite. Fig.1  
 157 graphically plots the variation of  $P_{el}$  and  $P_{eu}$  with  $V_1$ . The percolation threshold of the Maxwell  
 158 model is  $V_p=1$  when Eq.(1) is used and  $V_p=0$  when Eq.(2) is used. Eqs.(1) and (2) represent the  
 159 two extreme cases of mixes in the two-phase composite. In reality, however, neither the medium  
 160 nor inclusions would be perfectly covered by the medium or inclusions. In general, Eq.(1) is  
 161 more suitable for the composite with small volume fraction of inclusions; whereas Eq.(2) is  
 162 more apposite for the composite with large volume fraction of inclusions. Therefore, with the  
 163 increase in volume fraction of inclusions, the effective property of the composite will have a  
 164 “jump” from the lower-bound curve to the upper-bound curve. The volume fraction  
 165 corresponding to the “jumping” point is characterised as the percolation threshold of the  
 166 composite. By assuming the medium and inclusions are perfectly symmetric in the geometry of  
 167 the composite, Bruggeman [30] and Landauer [31] proposed independently an effective  
 168 medium approximation for calculating the effective property of the two-phase composite,  
 169



170



Fig.1 Effective property in Maxwell and Bruggeman-Landauer models ( $P_1=50P_2$ ).

$$\frac{(P_{es}-P_1)}{(2P_{es}+P_1)}V_1 + \frac{(P_{es}-P_2)}{(2P_{es}+P_2)}(1-V_1) = 0 \quad (3)$$

where  $P_{es}$  is the effective property of the two-phase composite. The variation of  $P_{es}$  with  $V_I$  described by Eq.(3) is also plotted in Fig.1. It can be seen from the figure that  $P_{es}$  almost coincides with  $P_{el}$  for small volume fraction of inclusions and to  $P_{eu}$  for large volume fraction of inclusions. In the middle region of  $0.2 \leq V_I \leq 0.8$   $P_{es}$  drives away from the lower-bound curve and gradually approaches to the upper-bound curve. For the case of  $P_1 \gg P_2$ , Eq.(3) can be simplified as,

$$P_{es} = \begin{cases} P_2 & V_1 \leq 1/3 \\ P_2 + \frac{(3V_1-1)}{2}(P_1 - P_2) & V_1 > 1/3 \end{cases} \quad (4)$$

This indicates that the percolation threshold of the effective medium approximation is about 1/3, as shown by the black dot line in Fig.1. Eq.(3) illustrates how the volume fraction of inclusions influences  $P_{es}$ , leading it to vary from lower-bound curve to upper-bound curve.

Here, however, it should be noted that Eq.(3) was developed for spherical inclusions in which the aspect ratio of inclusions is one. For many fibre-reinforced composites the aspect ratio of inclusions is much greater than one [3,21] and the corresponding percolation threshold would be very low. To take account for the effects of aspect ratio and percolation threshold of inclusions on the effective properties of fibre-reinforced composites, we have to know the relationship between the percolation threshold and aspect ratio of inclusions and find out where the effective property leaves from the lower-bound curve and how quickly it reaches to the upper-bound curve.

### 3. Percolation threshold of fibre-reinforced two-phase composite materials

195

196 For a given two-phase composite material, if the aspect ratio and mixed pattern of inclusions  
197 in the composite are known then the percolation threshold and corresponding effective  
198 properties of the composite can be calculated analytically or numerically. For example, Pan et  
199 al. [2] presented an analytical expression of percolation threshold for transversely orthotropic  
200 composites containing randomly oriented ellipsoidal inclusions. Fang et al. [21] proposed an  
201 analytical expression of percolation threshold for isotropic composites containing prismatic  
202 inclusions equally distributed in three axial directions. Both models showed that the percolation  
203 threshold decreases with increased aspect ratio of inclusions. However, the downside of these  
204 models is that when the aspect ratio tends to one the percolation threshold also attends to one.  
205 To improve the model proposed by Fang et al. [21], here we consider a unit cube as the  
206 representative volume element (RVE) of the fibre reinforced two-phase composite material.  
207 The volume of fibres in the RVE is represented by three identical ellipsoids perpendicularly  
208 placed along with three axes as shown in Fig.2a, which can be expressed as follows,

$$209 \quad \Omega_{in}(\lambda, a) = \frac{4\pi a^3}{\sqrt{(1+\lambda^2)^3}} \left( \frac{3-4\sqrt{2}}{3} + \frac{\sqrt{(1+\lambda^2)^3-1}}{\lambda^2} \right) \quad (5)$$

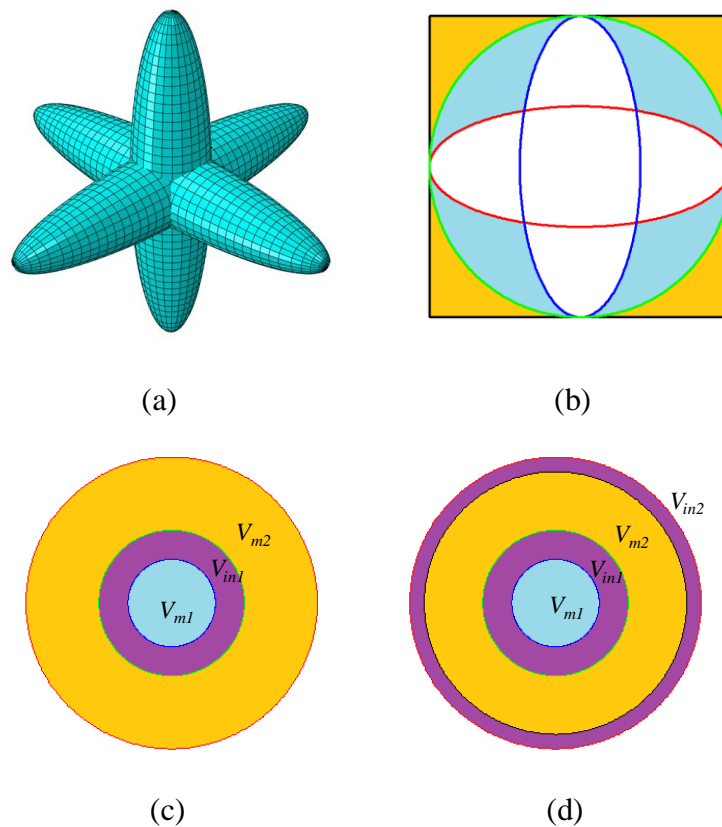
210 where  $\Omega_{in}$  is the volume of fibres in the RVE,  $\lambda$  is the aspect ratio of fibres, and  $a$  is the longer  
211 semi-axis of the ellipsoids. The detail of the derivation of Eq.(5) is given in the Appendix. At  
212 the state that the fibres reach to percolation, the inner ellipsoids should be inscribed in the outer  
213 cube as shown in Fig.2b. Mathematically this condition is expressed by  $2a=1$ . Thus, the  
214 threshold volume fraction of fibres, or the percolation threshold of the fibre-reinforced two-  
215 phase composite material, can be expressed as follows,

$$216 \quad V_p(\lambda) = \frac{\pi}{2\sqrt{(1+\lambda^2)^3}} \left( \frac{3-4\sqrt{2}}{3} + \frac{\sqrt{(1+\lambda^2)^3-1}}{\lambda^2} \right) \quad (6)$$

217 Note that if  $\lambda=1$  Eq.(6) gives  $V_p=\pi/6$ , indicating that the percolation threshold for the spherical  
218 inclusions in the present model is slightly greater than 0.5. The effective property of the

219 composite depends not only on the volume fraction of fibres but also on the mixed pattern of  
 220 fibres in the composite. The volume fraction may also affect the mixed pattern. Three different  
 221 mixed patterns are assumed in the present approach in terms of the volume fraction of fibres.  
 222 Case one is when the volume fraction of fibres is less than its percolation threshold  $V_p$ . In this  
 223 case the mixed pattern is assumed as a three-layer sphere as shown in Fig.2c, in which the core  
 224 part represents the matrix enclosed by the outer sphere inscribing the ellipsoids with partial  
 225 volume fraction  $V_{m1}$ , the middle layer represents the fibres with volume fraction  $V_{in1}$ , and the  
 226 outer layer represents the rest matrix with partial volume fraction  $V_{m2}$ . These three volume  
 227 fractions are defined as follows,

228



229

230

231

232

233 Fig.2. (a) Three ellipsoids placed along with three axes. (b) Definitions of volumes of fibres  
 234 and matrix in composite. Mixed patterns (c) before and (d) after percolation.

235

236 Case 1:  $V_{in} \leq V_p$

$$237 \quad V_{in1} = V_{in} = \frac{4\pi a^3}{\sqrt{(1+\lambda^2)^3}} \left( \frac{3-4\sqrt{2}}{3} + \frac{\sqrt{(1+\lambda^2)^3-1}}{\lambda^2} \right) \quad (7)$$

$$238 \quad V_{m1} = \frac{4\pi a^3}{3} - V_{in} \quad (8)$$

$$239 \quad V_{m2} = 1 - \frac{4\pi a^3}{3} \quad (9)$$

240 where  $V_{in}$  is the volume fraction of fibres in the composite. For given  $V_{in}$  and  $\lambda$ ,  $a$  can be  
 241 calculated from Eq.(7). For the two-layer sphere with core part (matrix) plus the middle layer  
 242 (fibres) we can use Eq.(2) by letting  $P_1=P_{in}$ ,  $P_2=P_m$ ,  $V_1=V_{in1}/(V_{in1}+V_{m1})$  to obtain its effective  
 243 property,

$$244 \quad P_e^{(0)} = P_{in} \frac{(P_m+2P_{in}) + \frac{2V_{m1}}{V_{in1}+V_{m1}}(P_m-P_{in})}{(P_m+2P_{in}) - \frac{V_{m1}}{V_{in1}+V_{m1}}(P_m-P_{in})} \quad (10)$$

245 where  $P_{in}$  and  $P_m$  is the property of the fibres and matrix, respectively. For the three-layer  
 246 sphere we can use Eq.(1) by letting  $P_1=P_e^{(0)}$ ,  $P_2=P_m$ ,  $V_1=(V_{in1}+V_{m1})/(V_{in1}+V_{m1}+V_{m2})$  to obtain  
 247 its effective property,

$$248 \quad P_e^{(1)} = P_m \frac{(P_e^{(0)}+2P_m) + \frac{2(V_{in1}+V_{m1})}{V_{in1}+V_{m1}+V_{m2}}(P_e^{(0)}-P_m)}{(P_e^{(0)}+2P_m) - \frac{V_{in1}+V_{m1}}{V_{in1}+V_{m1}+V_{m2}}(P_e^{(0)}-P_m)} \quad (11)$$

249 where  $P_e^{(1)}$  represents the effective property of the three-layer sphere with the mixed pattern as  
 250 shown in [Fig.2c](#), respectively.

251

252 Case two is when the volume fraction of fibres is greater than its percolation threshold  $V_p$ . In  
 253 this case the mixed pattern is assumed as a four-layer sphere as shown in [Fig.2d](#), in which the  
 254 core part represents the matrix enclosed by the outer sphere of unit diameter inscribing the  
 255 ellipsoids (see [Fig.2b](#)) with partial volume fraction  $V_{m1}=\pi/6-V_p$ , the inner middle layer  
 256 represents the part of the fibres with percolation volume fraction  $V_p$  (see [Fig.2b](#)), the outer  
 257 middle layer represents the rest matrix with partial volume fraction  $V_{m2}$ , and the outer layer

258 represents the rest fibres with partial volume fraction  $V_{in2}$ . These four volume fractions are  
 259 defined as follows,

260

261 Case 2:  $V_p \leq V_{in} \leq 1 - \pi/6 + V_p$

$$262 \quad V_{in1} = V_p \quad (12)$$

$$263 \quad V_{in2} = V_{in} - V_p \quad (13)$$

$$264 \quad V_{m1} = \frac{\pi}{6} - V_p \quad (14)$$

$$265 \quad V_{m2} = 1 - V_{m1} - V_{in} \quad (15)$$

266

267 The effective property of the four-layer sphere can be calculated as follows,

$$268 \quad P_e^{(2)} = P_{in} \frac{(P_e^{(1)} + 2P_{in}) + \frac{2(V_{in1} + V_{m1} + V_{m2})}{V_{in1} + V_{in2} + V_{m1} + V_{m2}} (P_e^{(1)} - P_{in})}{(P_e^{(1)} + 2P_{in}) - \frac{V_{in1} + V_{m1} + V_{m2}}{V_{in1} + V_{in2} + V_{m1} + V_{m2}} (P_e^{(1)} - P_{in})} \quad (16)$$

269 where  $P_e^{(1)}$  is defined by Eq.(11). The concept of using a recursive formula for Eqs.(10), (11)  
 270 and (16) is illustrated in [21,32] and thus is not discussed further here. Finally, case three is  
 271 when the volume fraction of fibres is greater than  $(1 - \pi/6 + V_p)$ , in which case the matrix  $V_{m2}$  in  
 272 the outer middle layer of the four-layer sphere disappears and thus the effective property of the  
 273 composite reaches to its upper-bound, that is,

274

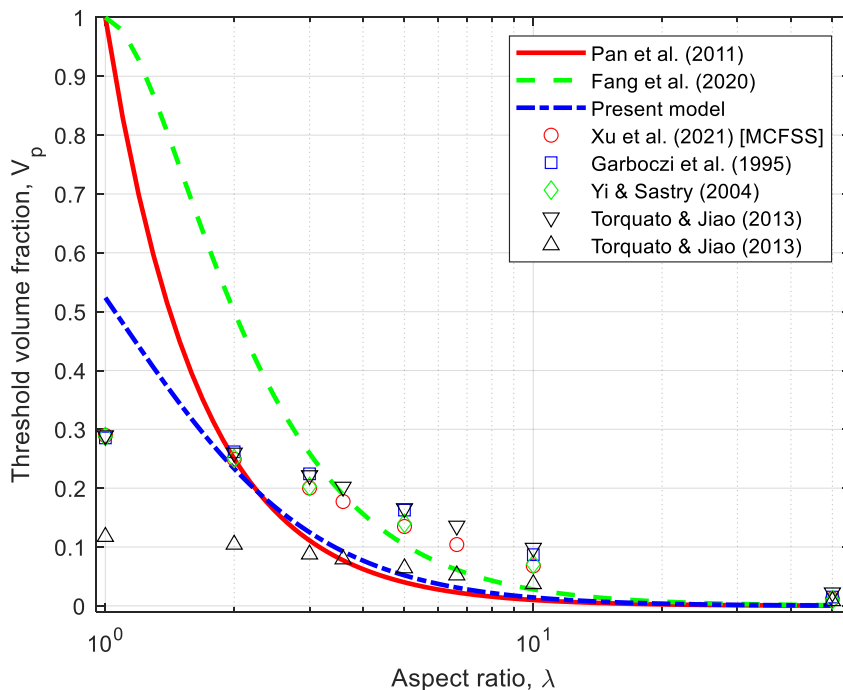
275 Case 3:  $1 - \pi/6 + V_p \leq V_{in}$

$$276 \quad P_e^{(3)} = P_{eu} \quad (17)$$

277

278 Physically,  $P_e^{(3)}$  represents the effective property of the sphere where the matrix is completely  
 279 enclosed by the fibres. Eqs.(11), (16) and (17) represent the non-percolated, partially  
 280 percolated, and fully percolated composite, respectively. The effective property calculated  
 281 using these equations reflect the effects of volume fraction, aspect ratio, and pattern of

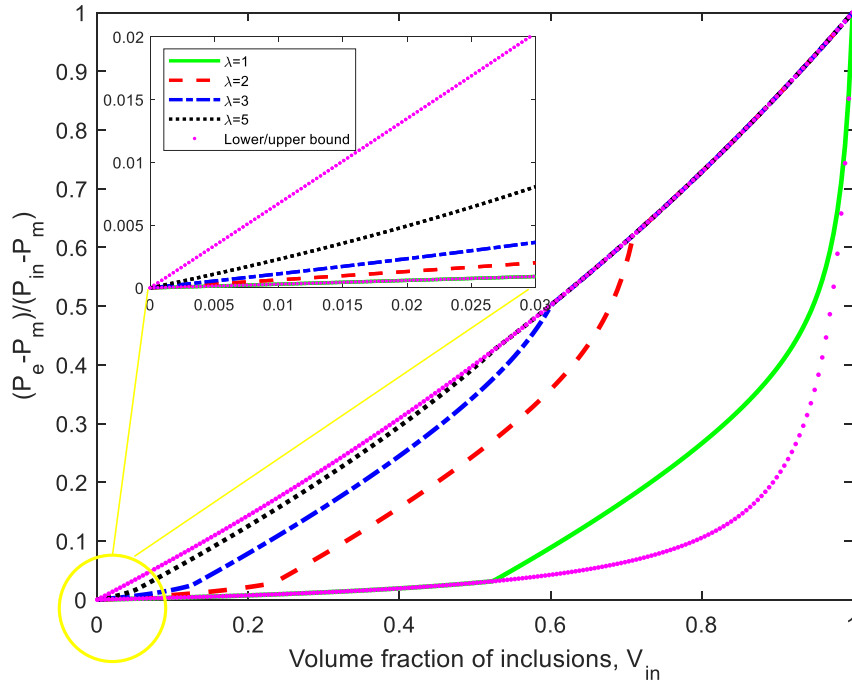
282 distribution and dispersion of fibres in the composite. The effective property of the composite  
 283 is calculated differently in different zones. The definition of the three zones is governed by the  
 284 percolation threshold. Fig.3 shows a comparison of the percolation threshold obtained from  
 285 various different models including analytical [2,21] and numerical [26,33,34,35] models. It can  
 286 be seen from the figure that the percolation thresholds given from different models are almost  
 287 identical when the aspect ratio is greater than 10. In the region of  $4 > \lambda > 8$ , Fang's model is  
 288 slightly closer to those predicted by numerical models; whereas the present model performs  
 289 reasonable well for all range of the aspect ratio. Fig.4 plots the variation of the effective property  
 290 with the volume fraction of fibres for different aspect ratios. The figure demonstrates that, the  
 291 effective property increases slowly in the non-percolated zone; then quickly in the partially  
 292 percolated zone, and finally it merges to the upper-bound curve in the fully percolated zone.  
 293 The aspect ratio of fibres has a significant effect on the positions where the effective property  
 294 starts to jump and where it lands to the upper-bound curve. However, the aspect ratio has no  
 295 influence on the effective property after it reaches to the upper-bound.  
 296



297

298 Fig.3 Variation of threshold volume fraction with aspect ratio of inclusions at percolation.

299



300

301 Fig.4 Variation of effective property with volume fraction of inclusions

302 for different aspect ratios ( $P_1=100P_2$ ).

303

#### 304 4. Validation of model with experimental results

305

306 To validate the present model, experimental data published in literature are used for three

307 different composite materials. The first one is the multi-walled CNTs-reinforced cement paste

308 for which the experimental data of effective electrical conductivity were reported in [36]. From

309 the simulation it is found that the percolation threshold of the CNTs-reinforced cement is about

310  $V_p=4.83\%$ . Thus the experimental data obtained for  $V_{in}<1.01\%$  are all in the non-percolated

311 zone. In the calculation, the value of  $P_m$  (electrical conductivity of cement paste) is obtained

312 directly from the test, whereas the value of  $P_{in}$  (electrical conductivity of CNTs) is reduced after

313 taking into account the tunnelling effect of CNTs. Fig.5 shows the comparison between the

314 model prediction and test data. It can be seen from the figure that the effective electrical  
315 conductivities predicted in the model and measured in the experiment are reasonably close and  
316 have very similar variation attendance.

317

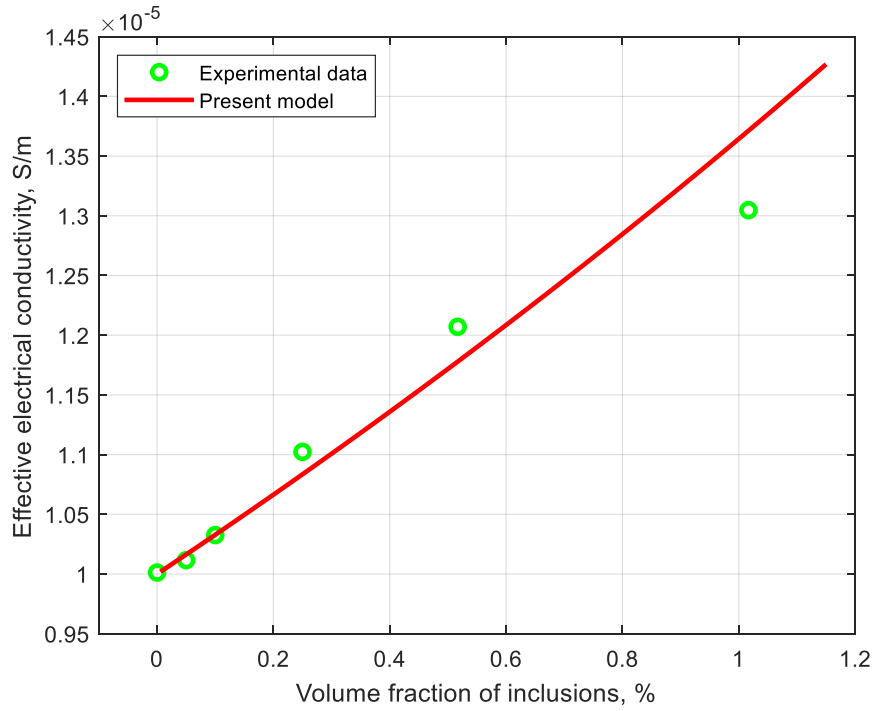
318 The second one is the moisture-enhanced cement paste for which the experimental data of  
319 effective electrical conductivity were reported in [36]. The percolation threshold of the  
320 moisture-enhanced cement is assumed to be  $V_p=0.5$ , which means that the pores in the cement  
321 are very close to spherical shape. The experimental data obtained from the test for  $V_{in}<0.4$  thus  
322 are all in non-percolated zone. In the calculation, both the values of  $P_m$  (electrical conductivity  
323 of cement paste) and  $P_{in}$  (electrical conductivity of moisture) are obtained directly from the test.  
324 Fig.6 shows the comparison between the model prediction and test data. It can be seen from the  
325 figure that there is very good agreement between the predicted and measured effective electrical  
326 conductivities.

327

328 The third one is the A4 multi-walled CNTs-reinforced epoxy composite. The experimental data  
329 of effective electrical conductivity were reported in [37]. The percolation threshold of the  
330 CNTs-reinforced epoxy composite is found to be about  $V_p=0.12\%$ , which is much lower than  
331 that of the CNTs-reinforced cement paste. The experimental data obtained from the tests for  
332  $V_{in}\leq 3.0\%$  thus covers all of three zones. In the calculation, the value of  $P_m$  (electrical  
333 conductivity of epoxy) is obtained directly from the test, whereas the value of  $P_{in}$  (electrical  
334 conductivity of CNTs) is reduced after taking into account the tunnelling effect of CNTs. Fig.7  
335 shows the comparison between the model prediction and test data. Excellent agreement between  
336 the prediction and test data is demonstrated.

337





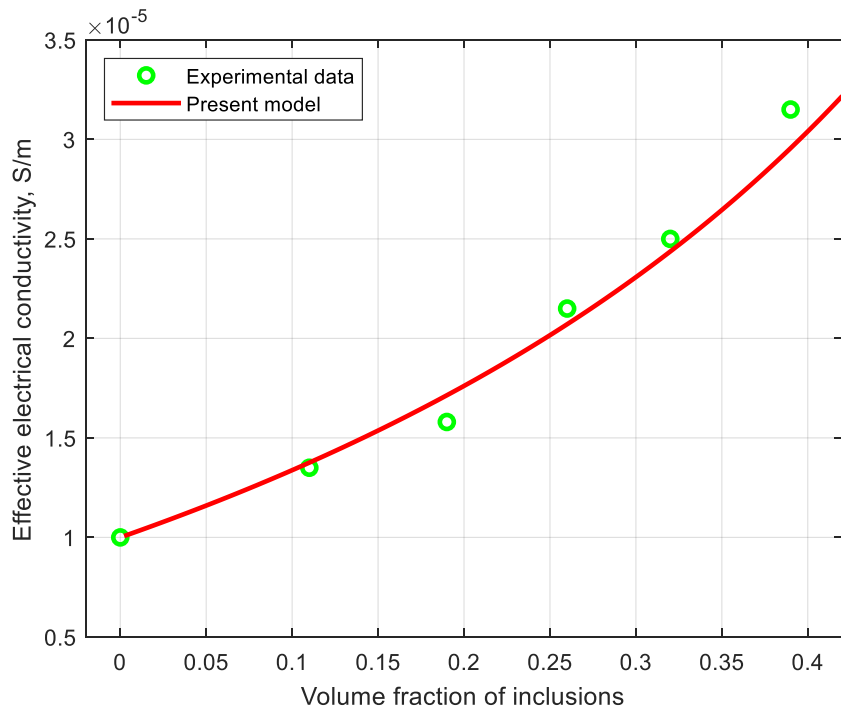
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339

Fig.5 Comparison of effective electrical conductivity of CNT-reinforced fresh mortar

340

between prediction and experiment ( $V_p=4.83\%$ ,  $P_m=1.0 \times 10^{-5}$  S/m,  $P_{in}=5.0$  S/m).



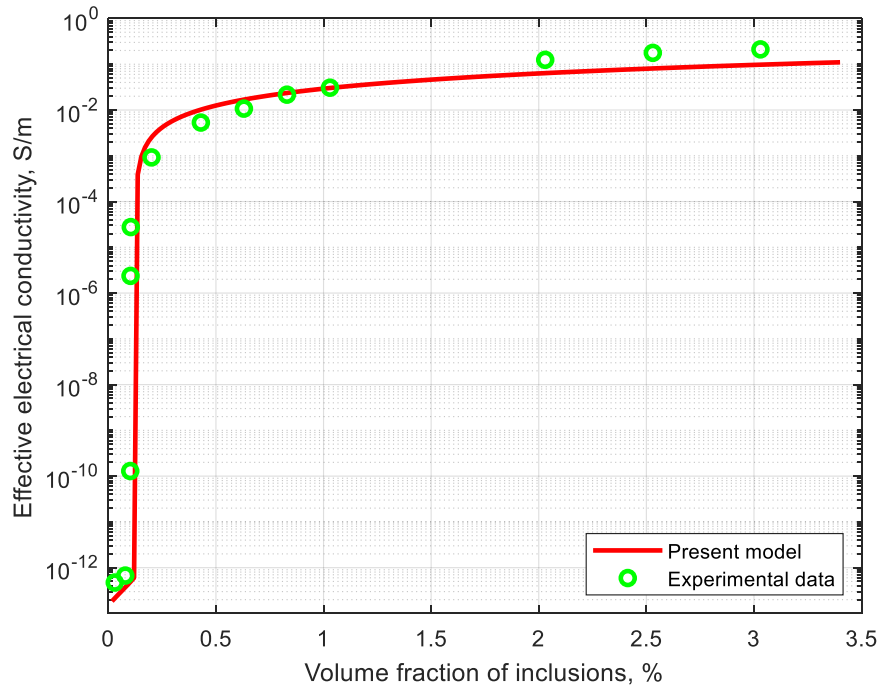
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342

Fig.6 Comparison of effective electrical conductivity of moisture-enhanced cement

343

between prediction and experiment ( $V_p=0.5$ ,  $P_m=1.0 \times 10^{-5}$  S/m,  $P_{in}=1.0 \times 10^{-3}$  S/m).



344

345 Fig.7 Comparison of effective electrical conductivity of CNT-reinforced epoxy composite

346 between prediction and experiment ( $V_p=0.12\%$ ,  $P_m=1.5 \times 10^{-13}$  S/m,  $P_{in}=5.0$  S/m).

347

348 **5. Conclusions**

349

350 This paper has presented an analytical model for determining percolation threshold and  
 351 calculating the effective properties of fibre reinforced two-phase composite materials. The  
 352 model has been validated by using experimental data published in literature. From the results  
 353 obtained we have the following conclusions.

- 354 • The percolation threshold of a two-phase composite can be determined by using a unit  
 355 length RVE cube inscribed by three equal ellipsoids placed perpendicularly along three  
 356 axes. The volume fraction of the ellipsoids inside the RVE represents the percolation  
 357 threshold of the composite.
- 358 • According to the percolation threshold, the effect of the volume fraction of inclusions  
 359 on the effective properties of the composite can be characterised in terms of non-

360 percolated zone, partially percolated zone, and fully percolated zone, respectively.

361 These three zones have different mixed patterns and thus the ways used to calculate the  
362 effective properties are also different.

363 • In the non-percolated zone, the inclusions infold a small part of matrix but themselves  
364 are completely implanted in the matrix. The change of volume fraction of inclusions in  
365 this zone leads to a limited change of effective properties of the composite due to the  
366 fact that the inclusions are isolated by the matrix.

367 • In the partially percolated zone, part of inclusions become connective although some of  
368 inclusions are still implanted in the matrix. With the increase of volume fraction of  
369 inclusions, more inclusions become connective, leading to a big change in the effective  
370 properties of the composite.

371 • In the fully percolated zone, all inclusions become connective and the matrix is  
372 completely enclosed by the connected inclusions. Thus the effective properties in this  
373 zone can be calculated by using the upper-bound formula of the two-phase composite.  
374 After it reaches to the upper-bound, the aspect ratio no longer has influence on the  
375 effective property of the composite.

376

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382 interests or personal relationships that could have appeared to influence the work reported in  
383 this paper.

384

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508

### 509 **Appendix: Calculation of volume of three overlapped ellipsoids**

510

511 The total volume of three overlapped identical ellipsoids shown in Fig.2a can be split into three  
 512 parts. One is the volume of the inner sphere with radius  $R_p$  as shown in Fig.A1. One is the  
 513 negative volume of six crowns of the sphere as shown in pink colour in Fig.A1. One is the  
 514 volume of six partial ellipsoids as shown in sky-blue colour in Fig.A1.

515

516 Assume the three semi-axes of the ellipsoid are  $(a, b, b)$ . Thus, the coordinates  $(x_p, y_p)$  and the  
 517 radius  $R_p$  of the inner sphere shown in Fig.A1 can be expressed as follows,

$$518 \quad x_p = y_p = \frac{ab}{\sqrt{a^2+b^2}} = \frac{a}{\sqrt{1+\lambda^2}} \quad (\text{A1})$$

$$519 \quad R_p = \sqrt{x_p^2 + y_p^2} = \frac{a\sqrt{2}}{\sqrt{1+\lambda^2}} \quad (\text{A2})$$

520 where  $\lambda=a/b$  is the aspect ratio of the ellipsoid. The volume of the inner sphere with radius  $R_p$   
 521 is calculated as follows,

$$522 \quad V_1 = \frac{4\pi}{3} R_p^3 = \frac{4\pi a^3}{3} \left( \frac{2}{1+\lambda^2} \right)^{3/2} \quad (\text{A3})$$

523 The negative volume of six crowns of the inner sphere is calculated as follows,

$$524 \quad V_2 = -6\pi(R_p - x_p)^2 \left( R_p - \frac{R_p - x_p}{3} \right) = -\frac{(8\sqrt{2}-10)\pi a^3}{(1+\lambda^2)^{3/2}} \quad (\text{A4})$$

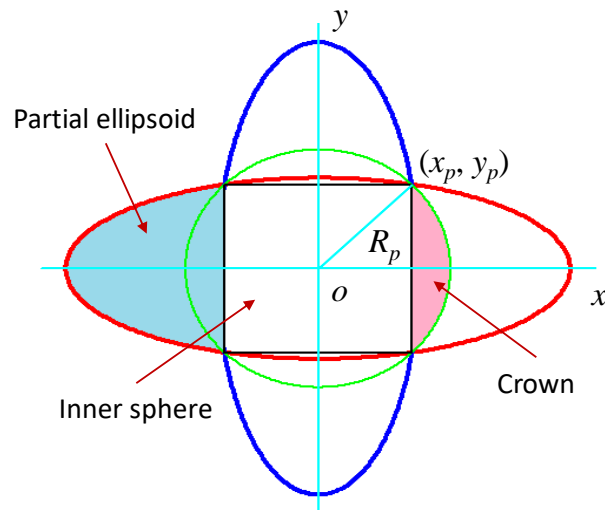
525 The volume of six partial ellipsoids is calculated as follows,

$$526 \quad V_3 = 6 \int_{x_p}^a \pi y^2 dx = \frac{2\pi a^3}{\lambda^2} \left( 2 - \frac{3x_p}{a} + \frac{x_p^3}{a^3} \right) = \frac{2\pi a^3}{\lambda^2} \left( 2 - \frac{3}{\sqrt{1+\lambda^2}} + \frac{1}{(1+\lambda^2)^{3/2}} \right) \quad (\text{A5})$$

527 Hence, the volume of the three overlapped ellipsoids is expressed as,

528 
$$\Omega_{in}(\lambda, a) = V_1 + V_2 + V_3 = \frac{4\pi a^3}{\sqrt{(1+\lambda^2)^3}} \left( \frac{3-4\sqrt{2}}{3} + \frac{\sqrt{(1+\lambda^2)^3-1}}{\lambda^2} \right) \quad (A6)$$

529 As expected, Eq.(A6) gives  $\Omega_{in}=4\pi a^3/3$  if  $\lambda=1$ . Also, it is indicated that if  $\lambda \rightarrow \infty$  then  $\Omega_{in} \rightarrow 0$ .  
 530 The former represents the spherical inclusions; whereas the latter stands for the line inclusions  
 531 of infinite small cross-section.  
 532



533  
 534 Fig.A1 Plan view of object of overlapped ellipsoids with semi-axes of  $(a, b, b)$ .